Inorganic Materials and Process for Bioresorbable Electronics

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ABSTRACT
This article highlights new opportunities of inorganic semiconductor materials for bio-implantable electronics, as a subset of ‘transient’ technology defined by an ability to physically dissolve, chemically degrade, or disintegrate in a controlled manner. Concepts of foundational materials for this area of technology with historical background start with the dissolution chemistry and reaction kinetics associated with hydrolysis of nanoscale silicon surface as a function of temperature and pH level. The following section covers biocompatibility of silicon, including related other semiconductor materials. Recent transient demonstrations of components and device levels for bioresorbable implantation enable the future direction of the transient electronics, as temporary implanters and other medical devices that provide important diagnosis and precisely personalized therapies. A final section outlines recent bioresorbable applications for sensing various biophysical parameters, monitoring electrophysiological activities, and delivering therapeutic signals in a programmed manner.

KEY WORDS
Bioresorbable, biodegradable, semiconductor, silicon fabrication, transient.

1. INTRODUCTION
Recent advances in chemistry of the materials introduced unusual classes of electronic devices, which dissolve, disintegrate, or otherwise physically or chemically decompose in a defined period. Such types of “transient” materials and processes may enable eco-electronics that minimize electronic wastes or provide extremely high security devices that disappear physically for military uses or intellectual property rights, which makes data recovery impossible [1], [2]. Another interesting opportunity of ‘transient’ electronics is an application to implantable bio-devices that disappear harmlessly in the body dissolve in adjacent tissues. This technology, sometimes referred to as bioresorbable electronics, provides diagnostic or therapeutic function, and then disappears entirely to eliminate the need for surgical removal [3], [4]. Flexible electronics in the early 2000s brought many opportunities in the new class of electronics, such as wearable gears and epidermal sensors. In the inorganic materials field, at that time, they were developing ultra-thin inorganic materials in a nanoscale (~300 nm) for high performance flexible electronics due to its stable mechanical and chemical structures, compared to those of organic materials [5]. The research of thinning silicon (Si) semiconductor up to 200 nm encountered dissolvable characteristics of monocrystalline Si in the water [1]. The nanoscale forms of Si that hydrolyze in biofluids to yield biocompatible byproducts over biologically relevant timescales qualitatively advanced the field by providing direct routes to high performance operation [6]. When combined with bioresorbable conductors, dielectrics, substrates, and encapsulation layers, this nanoscale Si layer can provide the basis for a broad,
general class of bioresorbable electronic technology. Other properties of Si, such as its piezoresistivity and photovoltaic capacity, allow other types of bioresorbable devices such as implantable photodetectors, strain gauges and pH sensors [3]. Additional choices in semiconductors (silicon germanium (SiGe), zinc oxide (ZnO), synthetic polymers) also enable versatile applications [7]. The most advanced bioresorbable devices now exist as complete systems with successful demonstrations of clinically relevant modes of operation in animal models [3], [4], [8].

This article highlights the concepts of the foundational materials for this transient technology, starting with the dissolution chemistry and reaction kinetics associated with the hydrolysis of nanoscale Si layer (doping concentration, crystallinity) as a function of temperature and pH. The following discussion focuses on key supporting materials, including a range of dielectrics, metals, and substrates with exhaustive biocompatibility tests. A final section summarizes bioresorbable electronics for sensing various biophysical parameters, monitoring electrophysiological activities in a programmed manner.

2. DISSOLUTION CHEMISTRY OF SILICON NANOMATERIALS

Monocrystalline, electronic-grade Si is generally considered chemically stable in ambient, aqueous environments, due to the spontaneous formation of a native oxide on its surface. This notion of stability depends, however, on structural dimensions and on observational timescales. Specifically, loss of material from the surface of a bulk Si wafer (thickness of ~1 mm) immersed in water at rates of a few nanometers per day can, in most cases, be neglected over laboratory timescales; such rates of loss for Si nanowires, nanoribbons or nanomembranes will lead to their complete disappearance during similar timeframes [1]. In the context of systems described in this review, active silicon nanostructures dissolve in relevant aqueous environments within several days or weeks, depending on the geometry, the chemical termination of the silicon surface, the type and level of doping of the silicon and the composition and temperature of the surrounding solutions.

Previous studies of Si surfaces with water molecules suggest that the most important reaction is hydrolysis in water, which generates orthosilicic acid (Si(OH)₄) and hydrogen as byproducts (Si + 4H₂O → Si(OH)₄ + 2H₂), as illustrated in Figure 1a. This chemical product (Si(OH)₄) presents in natural and biological fluids in the body, commonly at concentrations between 100 to 101 ppm [9]. Dissolution of silicon nanomaterials, typically ranges with μg-scale in total mass for biomedical applications. Specifically, the combined use of Si nanomaterials with water-soluble dielectric and conductive materials offer many opportunities. Figure 1b shows the representative results of hydrolysis of Si nanostructure...
(3 μm×3 μm×70 nm, p-type, 10-20 Ω·cm) in simulated biofluid (phosphate buffered saline, pH 7.4, 37 °C). Here, the dissolution rate is ~5 nm/day, which results in the complete disappearance of this nanoscale Si within 12-13 days [1]. Figure 1c summarizes the dependence of the reaction rate on the pH level from 6 to 14. Results from the high pH part of this range connect to the well-established kinetics of etching and micromachining of bulk silicon in alkaline solutions. The linear dependence of the thickness (h) on reaction time indicates that the chemistry involves surface erosion, without a significant contribution from reactive diffusion of water into the silicon. As a result, the simple expression h=h₀−Rt, where h₀ and R are the initial thickness and the reaction rate, respectively, captures the behaviors.
The biochemical complexity increases considerably upon the introduction of ions into the surrounding aqueous solutions. Certain ions significantly accelerate the rates of reaction [10], including those (e.g. Na⁺, Ca²⁺, Mg²⁺ cations and Cl⁻, HCO₃⁻, HPO₄⁻ anions) that are common in environmental and
Specifically, measurements of ellipsometry show oxide layer during the doping procedure [12,13]. (p-type) dopants at concentrations from $10^{17}$ cm$^{-3}$ to $[10]$. Significantly at $10^{20}$ cm$^{-3}$ with similar behavior concentrations (e.g. 1 M K$_2$HPO$_4$/KH$_2$PO$_4$). These the rate increases to 65 nm/day in high ion concentrations up to $10^{19}$ cm$^{-3}$, and then decrease trends. The dissolution rates remain the same for biological fluids at concentrations from 0.1 to $> 50$ g/L. In the case of low ionic concentration, the Si nanomaterials exhibit comparably low dissolution rate at $\sim $1 nm/day (e.g. 0.05 M K$_2$HPO$_4$/KH$_2$PO$_4$), while the rate increases to 65 nm/day in high ion concentrations (e.g. 1 M K$_2$HPO$_4$/KH$_2$PO$_4$). These variations arise from catalyzing effects of the ions, as suggested by density functional theory (DFT) for the case of anions (HPO$_4^{2-}$ or Cl$^-$) where interactions with surface Si atoms can weaken nearby Si-Si backbonds [10].

The rates of dissolution of nanoscale Si with boron (p-type) dopants at concentrations from $10^{17}$ cm$^{-3}$ to $10^{20}$ cm$^{-3}$ in simulated biofluid (phosphate buffer solution, 0.1 M, pH 7.4) show that the dopants in the silicon also affect the reaction chemistry (Figure 1d). The dissolution rates remain the same for concentrations up to $10^{19}$ cm$^{-3}$, and then decrease significantly at $10^{20}$ cm$^{-3}$ with similar behavior resulting from doping with phosphorus (n-type) [11]. Such effects are well known in silicon KOH (10-57%) etching process with different quality of the native oxide layer during the doping procedure [12,13]. Specifically, measurements of ellipsometry show dense layers (~1 nm) of SiO$_2$ on highly doped Si (either boron or phosphorous) when immersed in water, possibly due to barrier-less oxidation pathways facilitated by the presence of dopants or lattice strains induced by the different atomic sizes of Si and dopants [12], [13]. On the other hand, lightly doped Si exhibits porous surface layers of SiO$_2$ that grow in parallel with the dissolution of Si. The rate of dissolution of SiO$_2$ is much slower than that of Si, thereby providing a plausible chemical explanation for the variation in rate with the doping level.

The crystallinity of Si affects the dissolution rate in water [14]. For example, polycrystalline (poly-Si) and amorphous (a-Si) silicon are useful as active materials in thin solar cells and photodetectors; silicon germanium (SiGe) enables high-speed transistors. Figure 1e summarizes the dissolution kinetics of poly-Si, a-Si, Ge, and SiGe in phosphate buffer solution (pH 7.4, 0.1 M, 37 °C).

### 3. OTHER BIORESORBABLE ELECTRONIC MATERIALS

Recent works show that magnesium (Mg), zinc (Zn), tungsten (W), and molybdenum (Mo) can serve as biodegradable metals; systematic studies presents their dissolution behavior in thin-films and foils [15,16]. These metals react with water, according to: Mg + $2H_2O \rightarrow Mg(OH)_2 + H_2$, Zn + $2H_2O \rightarrow Zn(OH)_2 + H_2$, 2W + $2H_2O + 3O_2 \rightarrow 2WO_4^2-$, 2Mo + $2H_2O + 3O_2 \rightarrow 2H_2MoO_4$. Metallic alloys of these metals can be a possible option of tunable dissolution of transient electronics, such as AZ31B (Mg-Zn alloy). The dissolution rates of these metals and metal alloys (Mg, Zn, W, Mo, AZ31B) in a representative biofluid...
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(Hank’s solution) are $7 \times 10^{-2}$, $7 \times 10^{-3}$, $1.7 \sim 0.3 \times 10^{-3}$, $3 \times 10^{4}$, $2 \times 10^{-2}$ μm h$^{-1}$ at room temperature, respectively [16].
4. BIOCOMPATIBILITY OF SI NANOMATERIALS

The biocompatibility of Si nanomaterials and their byproducts of hydrolysis are critically important for biomedical applications. Figure 2a shows the typical poly(dimethylsiloxane) (PDMS)-based microincubation setup for in vitro assessment of the cytotoxicity. A patterned array of Si nanostructures dissolves in the PDMS well structure to allow the measurement of changes in the thicknesses of the Si nanostructure (Figure 2b). The arrays of square Si nanomaterials were no longer visible on the fourth day, indicating Si fully dissolved without significant cell death (Figure 2c). The main product, silicic acid, is a nontoxic small molecule that represents the most common form of bioavailable Si in the human body (serum contains 11–25 μg silicon/dL) [17]. The silicic acid does not accumulate within the body, but is absorbed by the adjacent tissue and is eliminated via the urinary pathway [18]. Separate studies of the biocompatibility of Si nanomaterials support similar conclusions [1, 3, 4, 6, 14]. Table 1 summarizes the representative results of bioresorbable materials or their combination of implantable device level. Hwang et al. [6] utilized metastatic breast cancer cell line (MDA-MB-231)) and reported viability remains above 93% throughout the complete dissolution process, showing no significant cell toxicity. Kang et al. [3] used mouse fibroblast cell (L929), showing >95% relative viability of all cells with no significant changes in the percentage of early or late apoptosis.

Analysis of mechanotransduction via traction force microscopy (TFM) can offer quantitate and direct monitoring of cellular activities [19]. Figure 2d highlights the microstructure of p-channel metal-oxide-semiconductor field-effect transistors (MOSFETs). As shown in Figure 2e, CCD18 cells incubated at 37 °C tend to adhere along structural features following previously reported topology effects. TFM analysis of CCD18 human colon fibroblasts cell represents the 10 mM concentration of silicic acid, which does not show cell toxicity without significant difference between groups (Figure 2f).

Animal model studies also support similar results of the cell cytotoxicity test. Si nanomaterials on silk substrates, as subdermal implants in mice, indicate no change of body weight and no additional generation of primary immune cells from the axillary and branchial draining lymph nodes [11]. Related biocompatibility studies involve Si nanomaterials with other transient materials, such as Mg, SiO2 and PLGA in the form of functional devices. In one case, pressure sensors implanted into the intracranial space of rat models for 2, 4, and 8 weeks, indicated no overt reactions of brain glial cells, and no focal aggregation of glial cells [3]. Astrocytosis (an increase in the number of astrocyte cells) and microglial activity at the cortical surface remain within normal limits, indicating a minimal immune reaction to the device or its byproducts [3]. Immunohistology of thin, flexible electronic systems for electrophysiological mapping on the surface of the brain indicates no significant astrogliosis and no microglial reaction throughout the functional lifetime of the device and its complete dissolution [4].

5. DEMONSTRATION IN THE TRANSIENT COMPONENTS AND DEVICES

Transient metal substrate offers robust, compatible with most commercial microelectromechanical system (MEMS) fabrication process, and thermally and chemically stable ways to build transient components. Figure 3a-c illustrates diodes, capacitor, and inductor on the biodegradable metal foils, such as Fe, Mo, W, and Zn (~10 μm thick). Nanoscale Si layer (~200 nm thick) with Mg electrodes (~300 nm thick) on a Zn foil (~10 μm thick) consists of an array of PIN diodes (Figure 3a, up), and shows typical electrical characteristics (Figure 3a, bottom). Capacitors (Figure 3b) and inductors (Figure 3c) are also possible. Parallel-plate capacitors of various lateral dimensions (left, black, 150 × 150 μm; red, 250 × 250 μm; blue, 400 × 400 μm; green, 550 × 550 μm) can be formed with SiO2 (~900 nm thick) as the dielectric and Mg as the electrodes on Mo foils (~10 μm). Capacitances (Figure 3b, bottom) in the frequency range between 0 and 25 GHz. Spiral inductors with two or three turns (Figure 3c) constructed using Mg traces for the electrodes and SiO2 (~900 nm thick) as the interlayer dielectric can be formed on Mo foils (~10 μm thick). Transient resistive memory and light-emitting diodes (LEDs) constructed with semiconductors and transient electrode are also possible. Figure 3d illustrates metal-insulator-metal structure to demonstrate a resistive memory device, in which W and Mg serve as inert and active electrodes, respectively, and silk fibroin film serves as the switching layer [20]. Figure 3e shows the I–V curves of a W/silk fibroin/Mg sandwich structure with swiped voltage from -4 to 4 V. The I–V characteristic after electroforming exhibits a typical bipolar resistive switching behavior. The narrow distributions of set voltage and reset voltage indicate versatility of the device. Reversible switching without degradation up to 100 cycles demonstrates possible transient memory application, as shown in Figure 3f. Figure 3g describes the structure of fully transient LEDs with oxygen vacancy doped n-type ZnO(001) films deposited on n-type Si(111) substrates by pulsed laser deposition. The threshold voltage of the ZnO LED is ~5 V, as observed in the optical power density–current–voltage (L–I–V) curve (Figure 3h), similar to typical ZnO p–n junction LEDs [21], 22]. The feature of broadband emission allows fabrication of red (R), green-yellow (G), and blue-violet (B) LEDs by the
addition of transient Fabry–Perot optical filters (Figure 3i).

6. FOUNDRY-COMPATIBLE APPROACHES FOR TRANSIENT ELECTRONICS

The efforts of foundry-based approaches for transient electronics can provide high potential to serve as the manufacturing basis for bioresorbable and implantable devices. For example, the recent portfolio of biodegradable materials, such as Si, SiO2, and W, with only minute amounts of nondegradable materials (Ti/TiN) allows transient CMOS fabrication [23]. X-FAB Semiconductor Foundries, which uses 1-μm design rules [24], [25], enable capacitors, p-channel and n-channel transistors on a 6-inch wafer, as shown in Figure 4a. The prepared transient components can be transfer-printed from 6-inch wafer after undercut release by TMAH anisotropic etching (Figure 4b). As illustrated in Figure 4c, the procedure for release from the substrate exploits lithographically defined structures and anisotropic wet chemical etching, with the BOX as a back-surface etch stop. Careful control, combined with optimized procedures for transfer printing, can lead to cumulative yields of greater than 99%. Schematic illustrations and SEM images of the undercut profiles appear in Figure 4d. The results are consistent with etching that proceeds in the Si <110> direction, bounded by (111) planes. By consequence, an undercut process that begins at the RIE trenches leaves {111} silicon pyramids positioned at the centers of the device blocks [23], [26]. The transfer printing process uses stamps of PDMS with relief features defined to manipulate one device at a time or large collections of them simultaneously (Figure 4e). Figure 4f-h show output voltage characteristics for a CMOS inverter (Figure 4f), a negative-AND (NAND) (Figure 4g), and a negative-OR (NOR) (Figure 4h) circuit obtained in this manner. For the inverter, the voltage transfer characteristics exhibit gains of up to ~21 at supply bias levels of ±3 V, with input low voltage (VIL) of 0.05 V and input high voltage (VIH) of 0.85 V, that are capable of a large noise margin for high noise immunity.

7. BIORESORBBLE ELECTRONIC IMPLANTS

Implantable devices offer diverse and essential functions for research, advanced diagnosis, and treatment. In certain cases, this function is most valuable for finite operation time, matched to intrinsic biological processes, such as wound healing. Here, removal of the devices after this operation time is required to eliminate the unnecessary load on the patient, and the associated risks of uncontrolled migration within the body, pathological tissue responses and infection. The surgical retrieval procedures, however, can involve complications themselves. Transient, bioresorbable electronics offer a potential solution in this context. The following summarizes recent research demonstrations.

Figure 5a shows a bioresorbable sensor of intracranial pressure, designed for monitoring recovery following a traumatic injury to the brain. The device incorporates a Si NM piezoresistive strain sensor integrated onto a flexible PLGA membrane that forms the top seal of an underlying cavity created on the etched region of the surface of a nanoscale Si substrate or Mg foil [3]. Changes in the pressure of surrounding fluids induce deflections of the membrane and changes in the resistance of the strain gauges, for electrical readout. Figure 5b illustrates its application as an intracranial pressure monitor with an interface to a wireless unit for data transmission. Here, degradable wires coated with bioresorbable polymers provide electrical connections between the sensor and external electronics. Studies using a rodent model indicate the stable measurement of temperature and pressure in the intracranial space for three days, and complete bioresorption within 4-6 weeks in biofluids at 37 °C [3].

Active electronic interfaces to the brain also have utility in this context of monitoring during recovery from an injury. Figure 5c shows passively and actively addressed systems for electrophysiological monitoring on the surface of the brain, where the constituent materials and fabrication strategies are similar to those for the devices in Figure 5a [4]. In these platforms, Si NM electrodes record normal physiologic and epileptiform activity such as electrocorticography (ECoG) and subdermal encephalograms (EEG), both in acute and chronic situations. Here, high doping levels in the Si electrodes lead to relatively slow rates of dissolution in biofluids, thereby enabling use for over 33 days. In addition to monitoring, bioresorbable electronic systems can also be utilized in various therapy modes [8]. For example, bi-layered radio frequency coils, built by patterned Mg film (~50 μm thick) and integrated with Si diodes (PIN diode, 200 nm thick), Mg capacitor (Mg/SiO2/Mg = 50-μm/600-nm/50-μm) can deliver therapeutic stimulation signal (20 Hz, monophasic pulsed wave) to the sciatic nerve in a rat model. The Mg coils, wirelessly powered by RF, allow full implantation in a back of rat, and the extended electrode can reach the target tissue (sciatic nerve), as shown in Figure 5e. The constituent materials completely disintegrate and dissolve within a month. Minimal inflammatory responses and fibrosis occur over an eight-week monitoring period, with no evidence of axonal injury or damage at the nerve-cuff interface. The application in the transected injured animal group shows enhanced therapeutic effects, such as increased muscle function reaction by measuring electromyography (EMG) signal. Furthermore, the
modified versions of this same platform can address a range of clinical scenarios and target tissues/organ systems, including the brain and spinal cord, skeletal muscles, and cardiac tissues with relatively few modifications to the form factor and the interfacial electrode sites.
8. SUMMARY AND PERSPECTIVE

Insights from the chemistry of dissolvable semiconductor materials open biodegradable classes of electronics, with broad potential utility in eco-electronics, military or industrial secure data management, and temporary implantable medical systems. The other materials in these platforms are equally important, and they offer many potential fields of research for future development in materials, fabrications, design, where high performance operation in sensing, actuation, and power harvesting. Additional opportunities lie in the development of passive, as well as actively stimulated, packaging materials, thereby allowing precisely controlled lifetimes of devices with a wide range of clinical options. The possible research directions, from fundamental studies of the semiconductor fabrication and surface chemistry to innovative engineering of the devices and systems, and the important potential applications in biomedicine suggest the future for this field.

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AUTHOR CONTRIBUTIONS

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REFERENCES


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